

# Materials Characterization: SEM/CL & XPS

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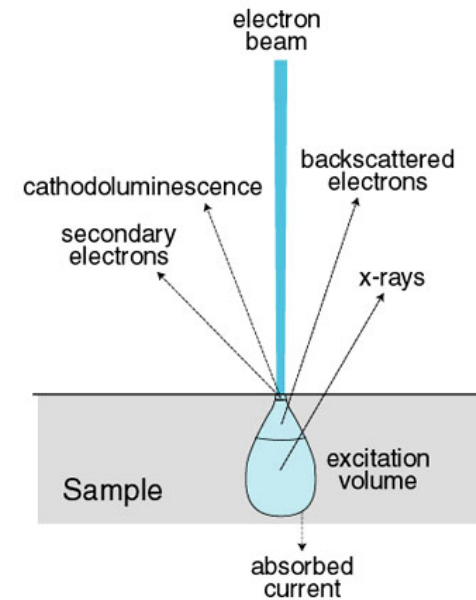
MURI Review Meeting  
May 2010



# SEM & CL

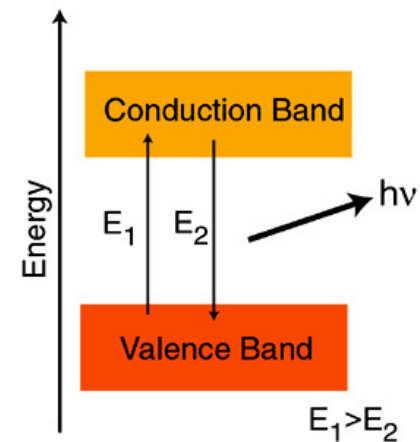
## Scanning Electron Microscope (SEM):

- Incident electron beam bombards sample
- Various electron and photon signals are generated
- Amount and type of signal depends on beam energy and excitation volume



## Cathodoluminescence (CL):

- Electron-hole pair recombination can result in both phonons and photons
- The photons generated are the basis of CL



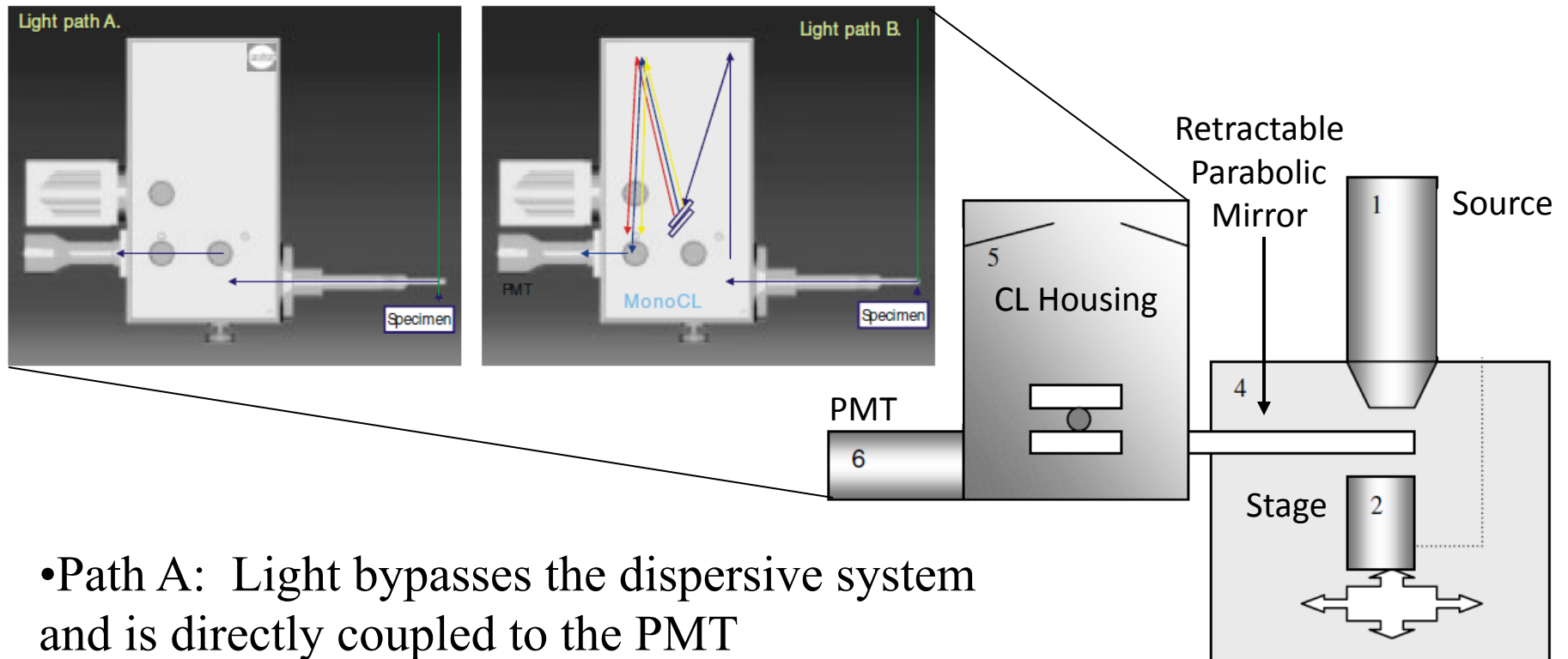
# SEM/CL Equipment

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FEI NanoSEM 430 with a Gatan MonoCL4 attachment



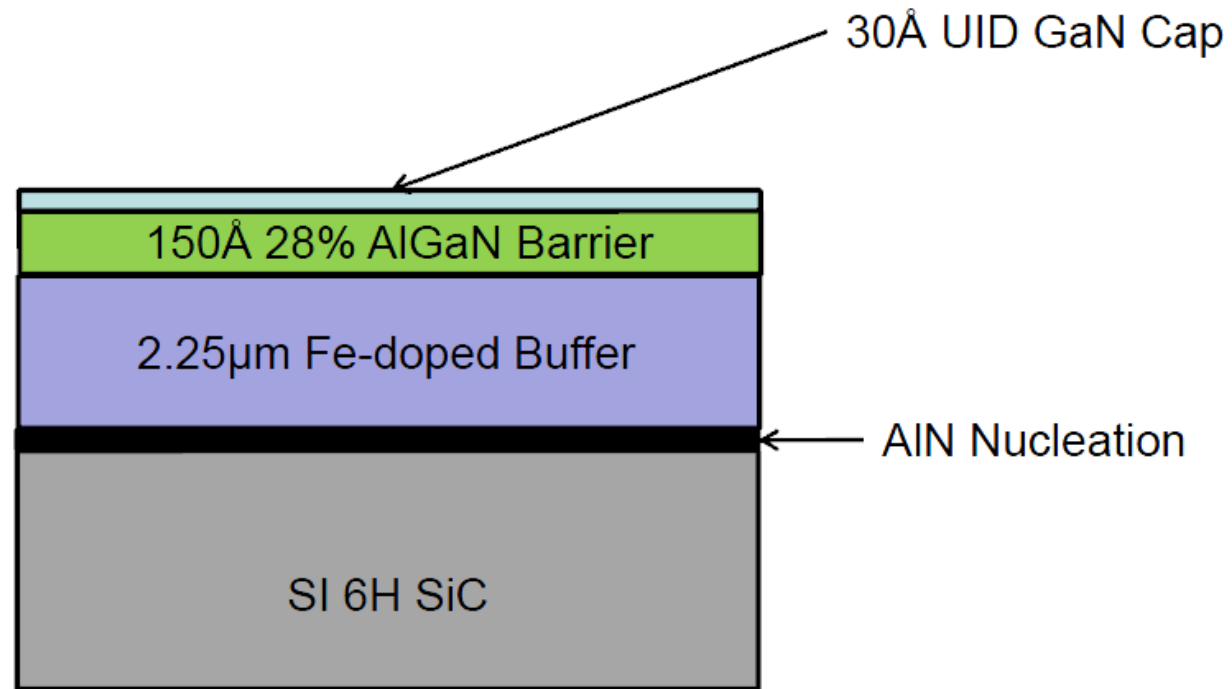
# Microscope Set-up & Beam Path



- Path A: Light bypasses the dispersive system and is directly coupled to the PMT
- Path B: Light is coupled through the dispersive system such that a chosen wavelength reaches the PMT

# HEMT Device Structure

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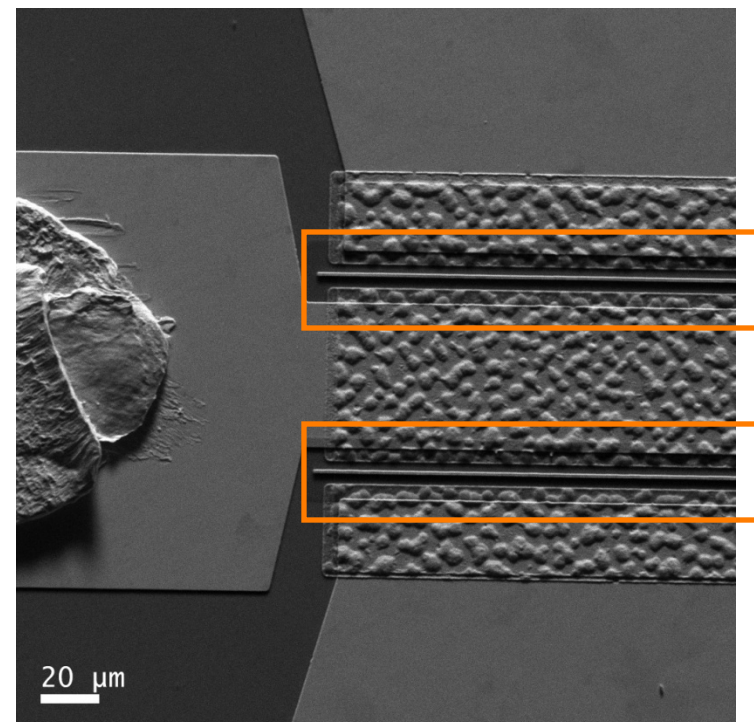


- Unstressed device
- Area analyzed includes T-gate and  $\text{SiN}_x$  passivation layer



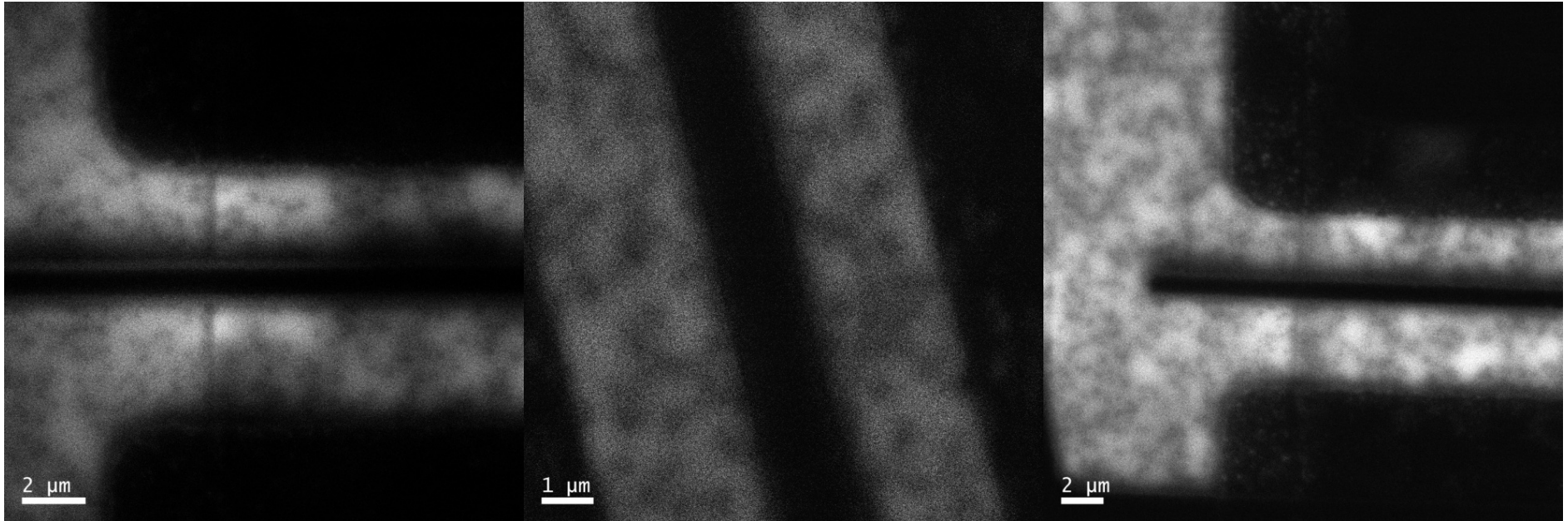
# SEM/CL Analysis Conditions

- “Non-destructive” analysis is done before and after failure
- Area of interest is in the channel/gate region
- Results shown are of depth dependent CL spectra
- Accelerating voltages were varied from 1 kV to 30 kV



# Device Failure CL Image Similarity

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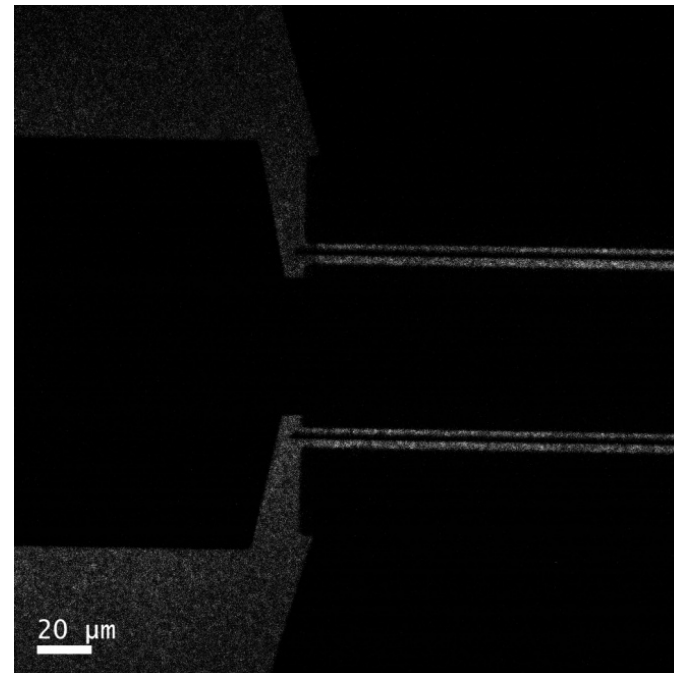
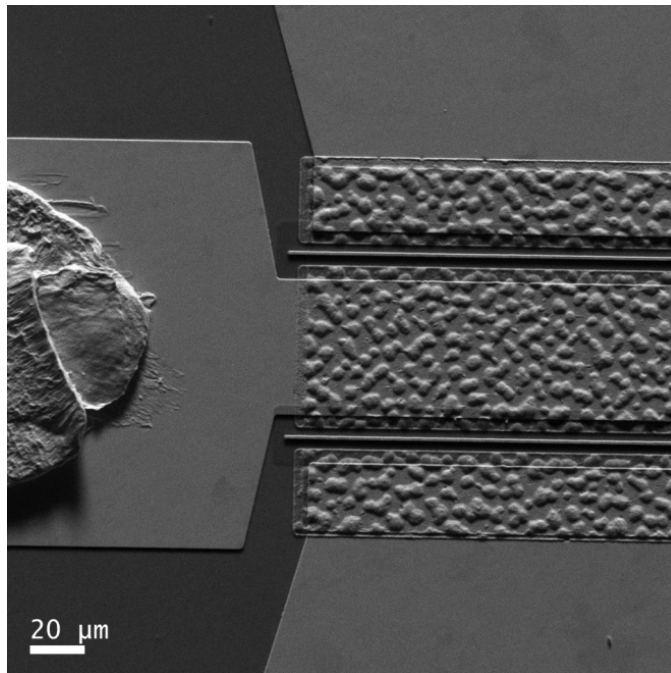


- Several devices stressed and failed in different ways all show the same defect structures (i.e. threading dislocations) due to similar contrast
- Accelerating voltages were too high and images are mainly representative of deep buffer layer

# SEM/CL Images

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Unstressed device was analyzed to form a baseline for microstructural defects for this study



15kV 1000x SE and CL images



# Electron Beam Interaction Volume

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- The penetration depth of the incident beam varies according to the following equation:

$$R(\mu m) = (0.052 / \rho) * E^{1.75}$$

With:  $\rho(GaN) = 6.1g / cm^3$

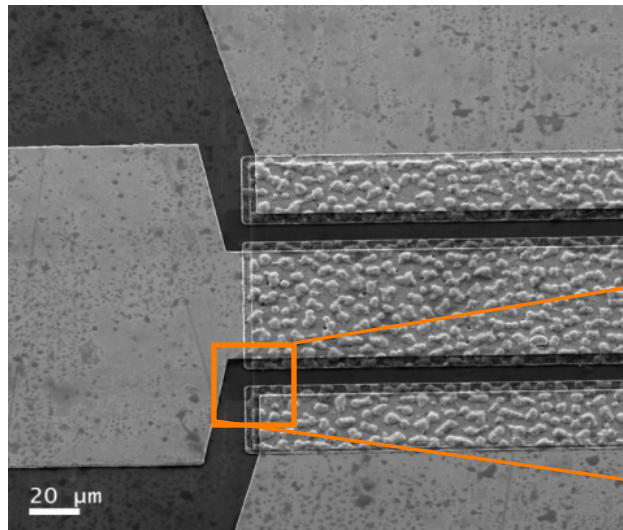
2 kV ~ 30 nm

5 kV ~ 200 nm

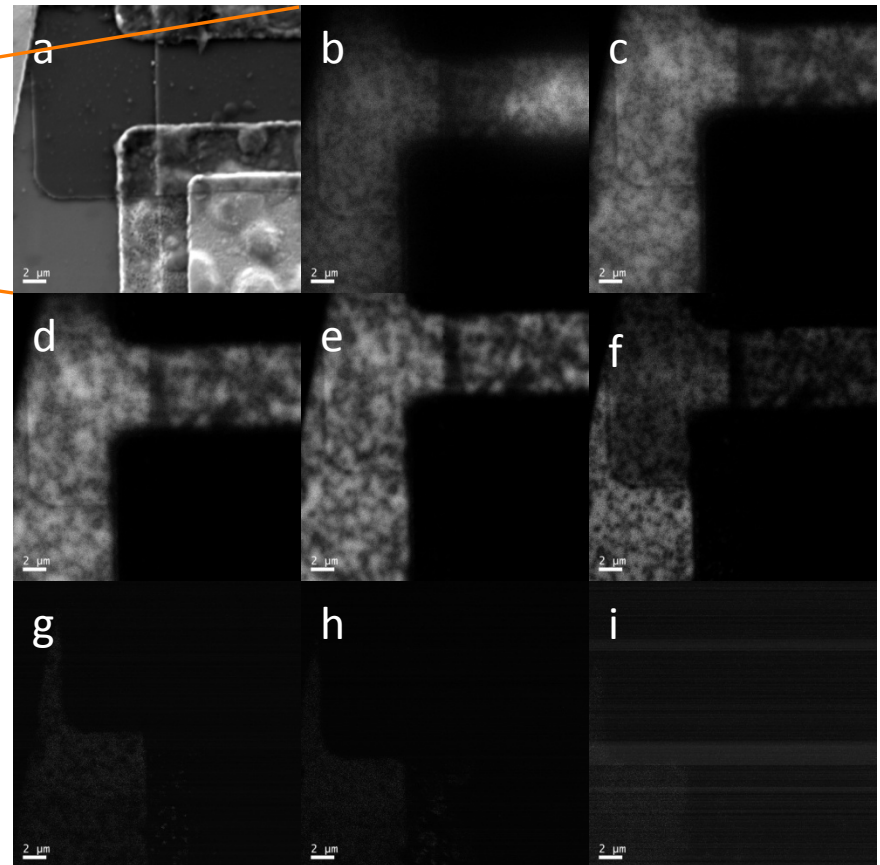
30 kV ~ 3200 nm

- The thickness of the passivation layer is ~200 nm

# Depth Dependence of CL Signal

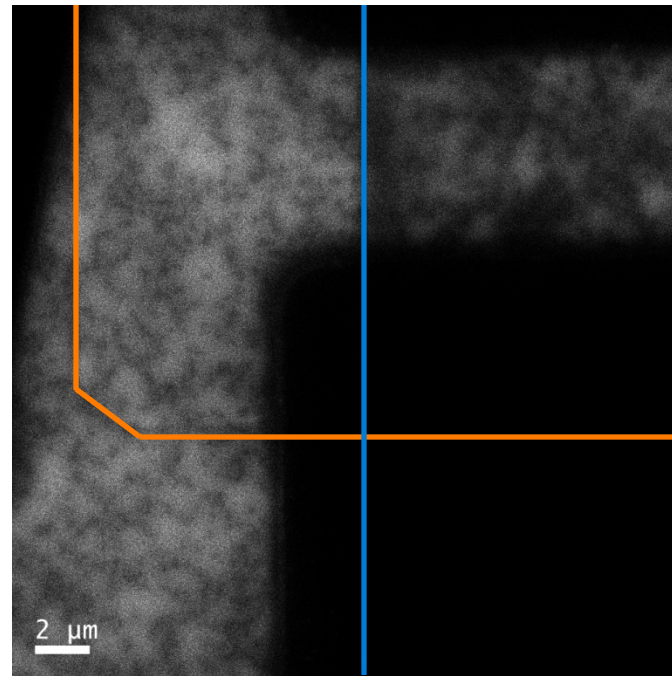
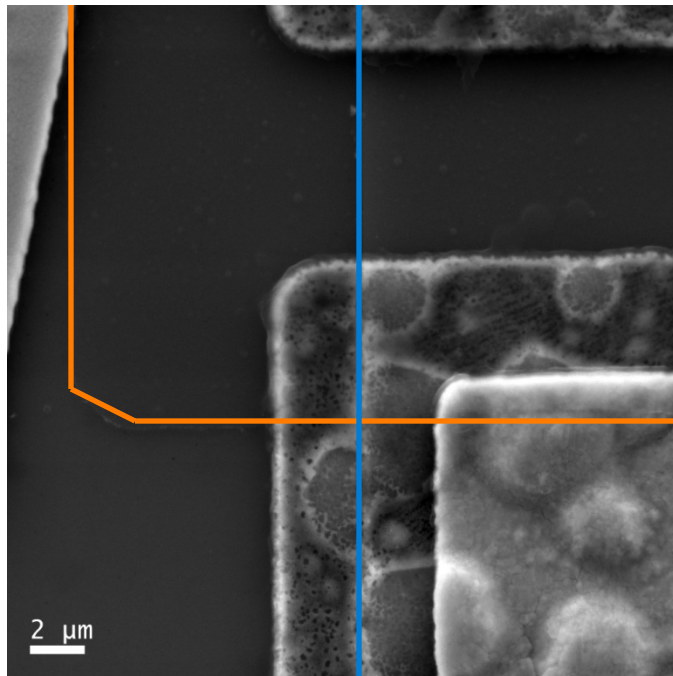


- a. SE image 10000x
- b. 30 kV CL image
- c. 20 kV CL Image
- d. 15 kV CL image
- e. 10 kV CL Image
- f. 5 kV CL image
- g. 3 kV CL image
- h. 2 kV CL image
- i. 1 kV CL image



# Substrate Defects

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20 kV 10000x SE and CL images

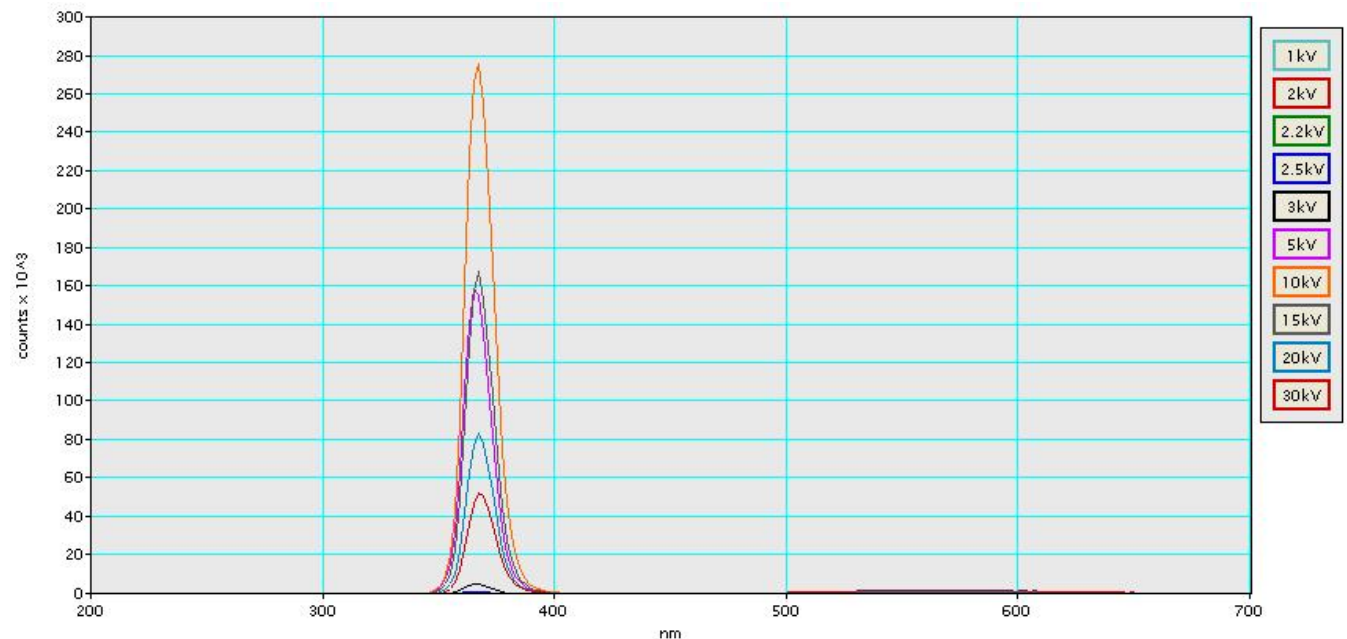
— SiN<sub>x</sub> passivation

— Mesa

# Monochromatic Spectroscopy

- Accelerating voltages:
  - Varied from 1 kV to 30 kV
  - Penetration depths from 8 nm to >3 mm
- Wavelengths scanned:
  - Spectrum from 200 to 700 nm

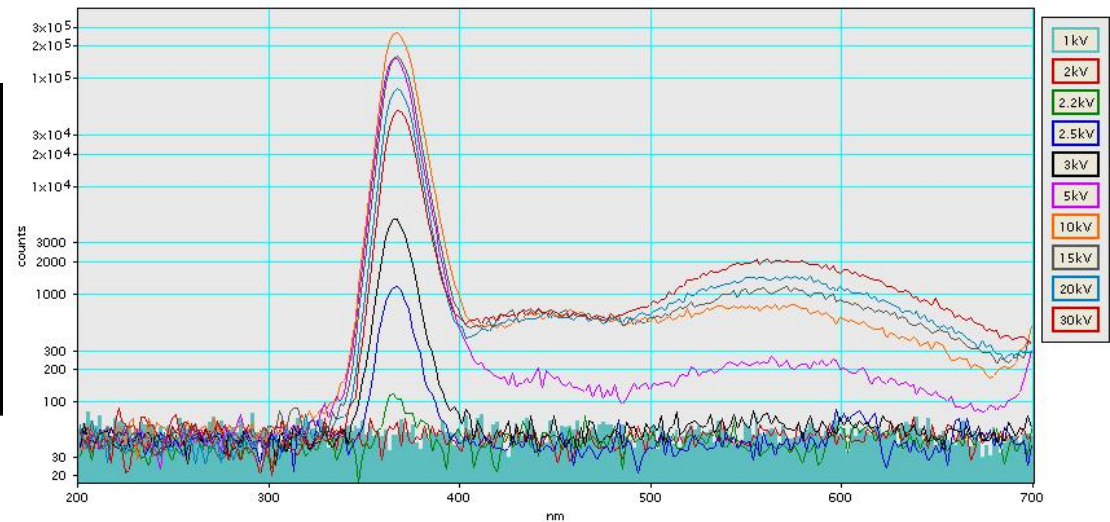
GaN band edge shows a maximum signal at 10 kV or ~366 nm





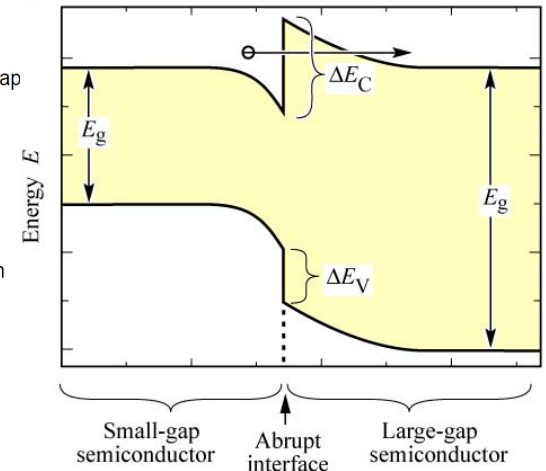
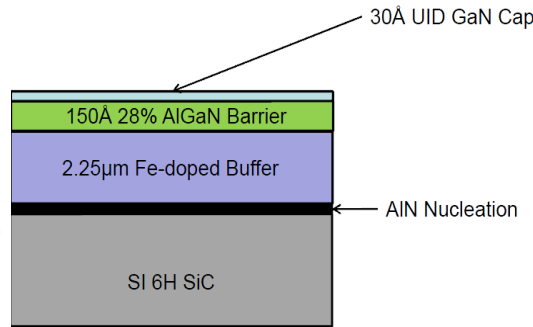
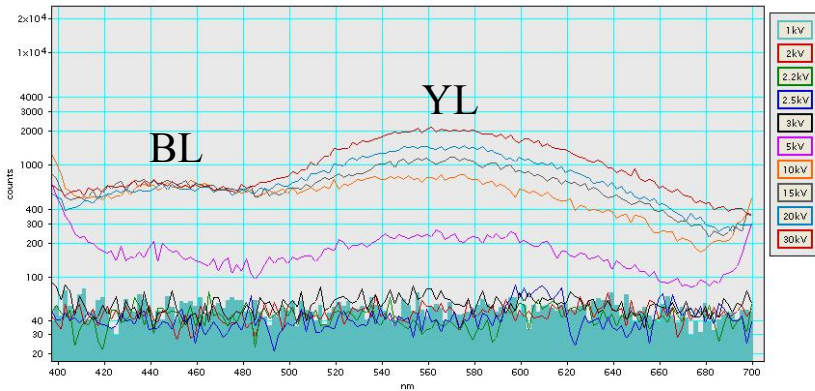
# Defect Peaks

	3 kV	5 kV	10 kV	30 kV
GaN band edge/BL	149.74	749.57	392.83	68.72
GaN band edge/YL	77.14	682.01	374.07	26.87
BL/YL	0.52	0.91	0.95	0.39



- Visible defect peaks:
  - Blue luminescence (BL) at ~440 nm
  - Yellow luminescence (YL) at ~556 nm
- The BL signal is stationary until we drop to 5 kV while the YL signal increases steadily with voltage

# BL & YL Peaks



BL:

- Disappears at 5 kV due to depth profiling within the top AlGaN/GaN layers
- The two most likely causes of the BL signal are:
  - $O_{Ga}$ -related defect
  - Fe from the doped GaN buffer layer

YL:

- Begins to appear at 5 kV and increases with depth (voltage) due to more defects from thicker buffer layer
- Typically associated with  $V_{Ga}$ -related defects

# SEM/CL Conclusions & Future Work

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## Conclusions:

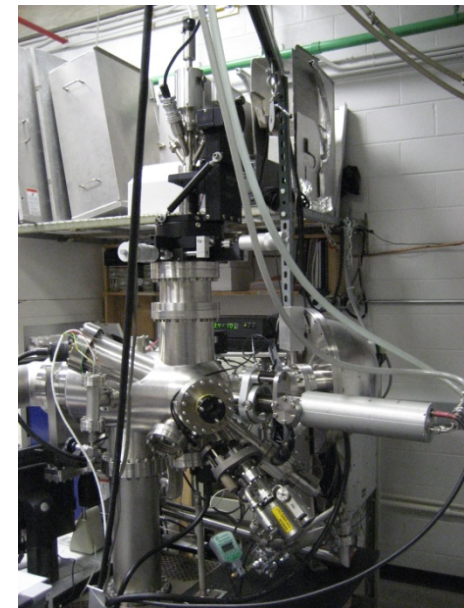
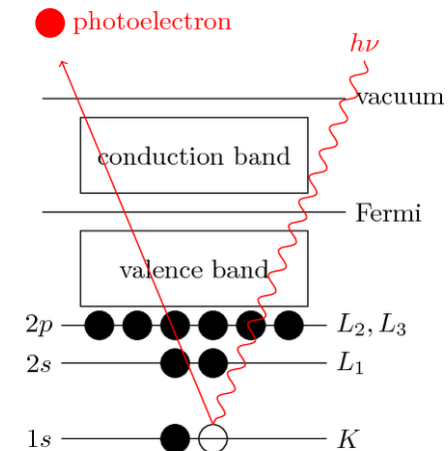
- Threading dislocations are present at all visible accelerating voltages and are unlikely to be sole contributor to YL and/or BL
- BL and YL are depth dependent
- Depth dependence and passivation layer thickness have significant effect on signal strength and wavelengths detected

## Future Work:

- Sample population of unstressed devices to be stressed will include stripping off passivation layer and gate to view the entire channel clearly through CL to determine cause of failure
- High resolution depth profiling

# X-ray Photoelectron Spectroscopy (XPS)

- AlGaN surface passivation:
  - Effects of UV ozone exposure time and XPS take-off angle
- Chemical changes of gate metal contact on GaN under device operation/stressing conditions
  - 300°C anneal in UHV to analyze material behavior under typical device operation conditions
  - 10 Å Ni deposited on GaN (Ga peak still visible)
  - 10 Å Pt deposited on GaN (Ga peak still visible)





# Previous Surface Passivation Studies

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SiN<sub>x</sub> on u-GaN:

- Shift in valence band (VB) of -0.4 eV and Ga 3d peak lost Ga-O portion after vacuum anneal at 200°C
- VB offset recovers to initial value after anneal in air at 200°C and Ga 3d peak recovered Ga-O portion
- Peak positions and relative intensities are nearly identical to pre-vacuum anneal values

SiN<sub>x</sub> on HEMT:

- Shift in VBM of -0.9eV after vacuum anneal and no determined Ga-O bonding in the Ga 3d peak
- VB offset recovers to initial value after air anneal, similar to SiNx/u-GaN sample

u-GaN:

- Shift in core level to VB only at anneal temp.
- Returns to pre-anneal value upon cool down
- Ga 3d peak shape consistent throughout anneal, no loss of Ga-O portion

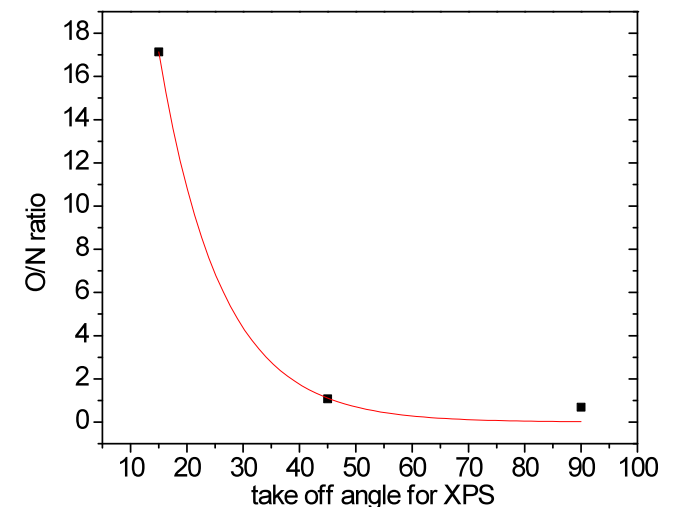
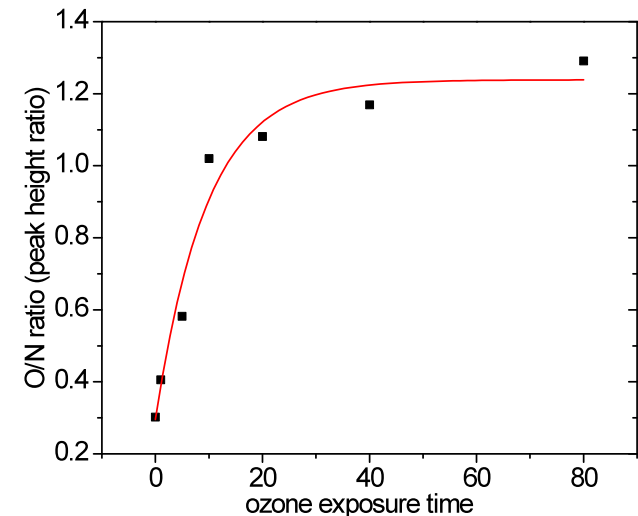
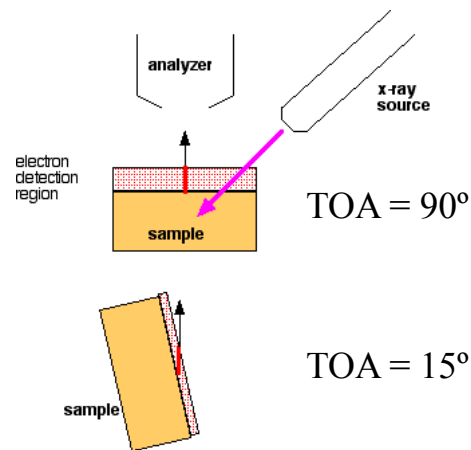
Different mechanism for the different samples due to the significant difference in VB shift.

# AlGaN Surface Passivation

Ozone effects on AlGaN:

- AlGaN samples placed in UV ozone oven, max. temp. from lamps  $\sim 50^{\circ}\text{C}$
- O/N ratio is surface dependent  $\rightarrow$  creation of oxynitride layer
- Future studies of  $\text{SiN}_x$  passivation layer after ozone exposure

Take-off angle (TOA):  
angle between energy  
analyzer and sample  
surface



# GaN Surface Treatment

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## Sample Preparation:

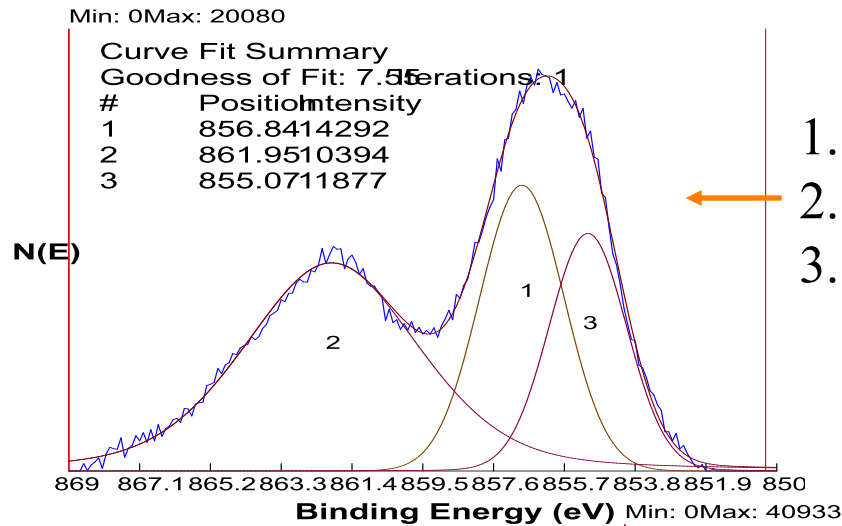
- Substrate: 3  $\mu\text{m}$  GaN on sapphire, 2" wafer
- Pretreat:
  - 3 min. 1:1 HCl:DI H<sub>2</sub>O, DI H<sub>2</sub>O rinse, N<sub>2</sub> dry
  - 25 min. UV ozone exposure
  - 5 min. BOE, DI H<sub>2</sub>O rinse, N<sub>2</sub> dry
- MBE anneal: 700°C for 30 min. in UHV
- Wafer cleaved into quarters:
  - One quarter for standard
  - One quarter 10 Å Ni metal deposition
  - One quarter 10 Å Pt metal deposition
  - One quarter set aside for future studies

# Ni on GaN Anneal Study

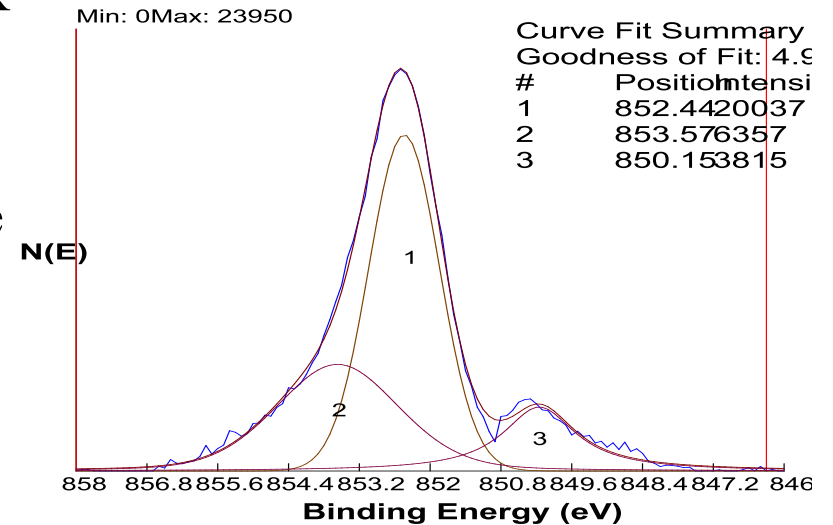
Pre-vacuum anneal

Ni  $2p_{3/2}$  Peak

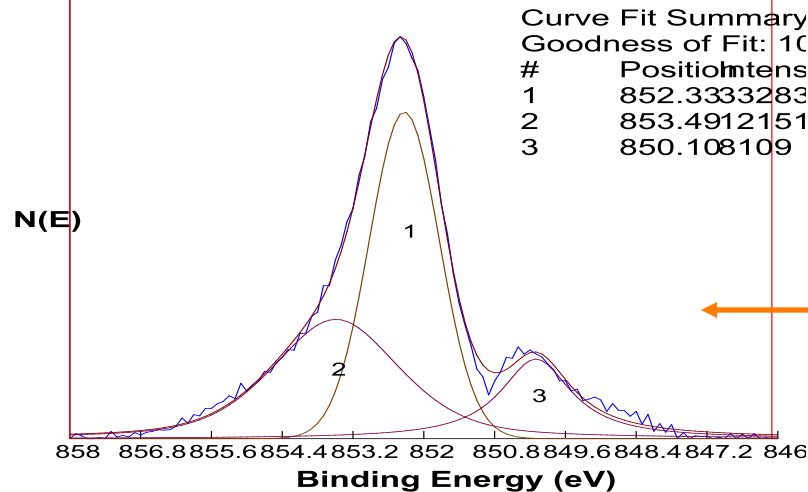
Post-vacuum anneal



Curves:  
1. Ni-O  
2. Ni satellite  
3. Ni metal



Vacuum  
anneal 300°C

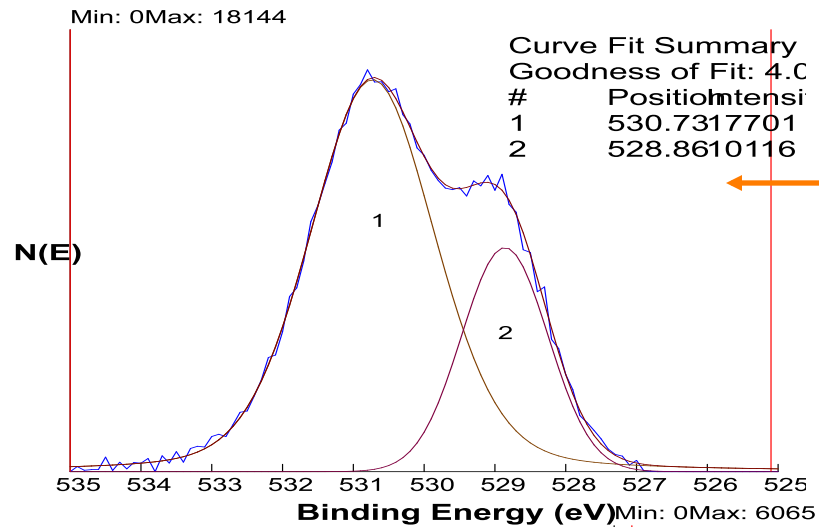


Curves:  
1. Ni-O  
2. Ni-O satellite  
3. Ni metal



# Ni on GaN Anneal Study

Pre-vacuum anneal

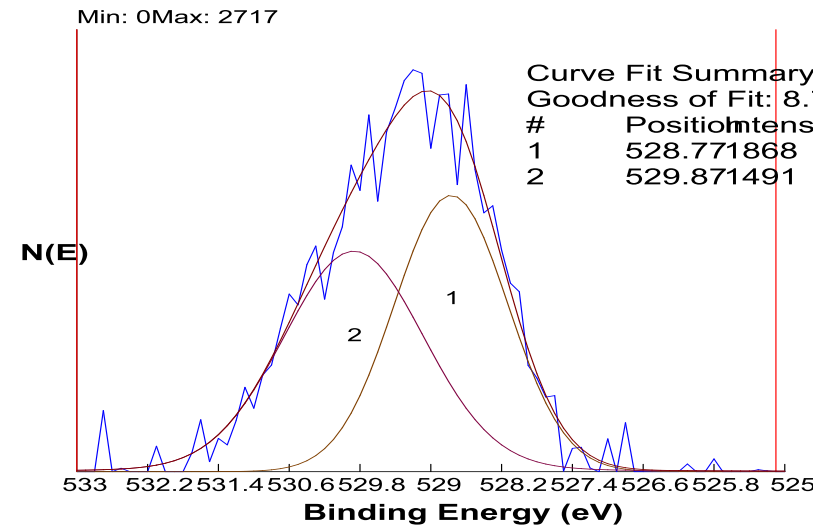


O 1s Peak

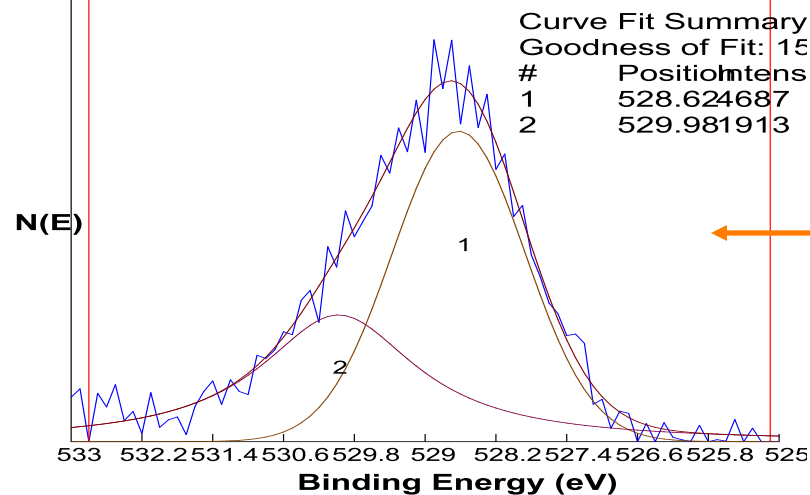
Curves:

1. Ga-O
2. Ni-O

Post-vacuum anneal



Vacuum  
anneal 300°C

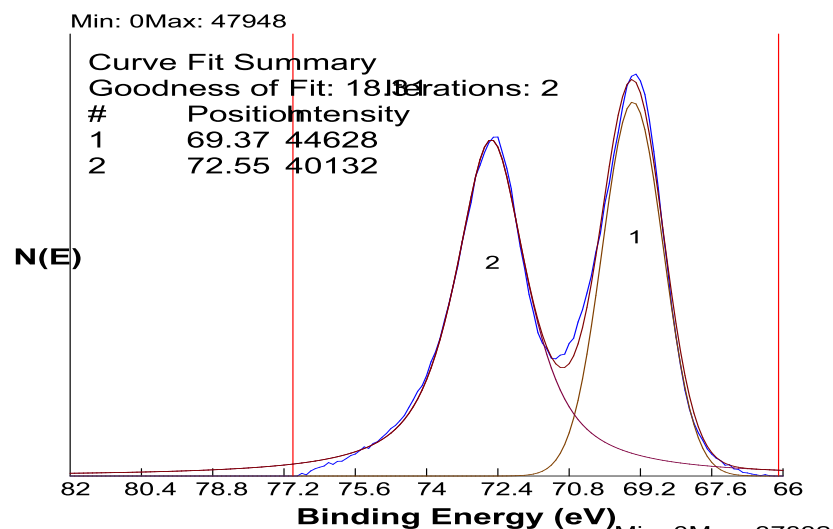


Curves:

1. Ni-O
2. Ga-O

# Pt on GaN Anneal Study

Pre-vacuum anneal

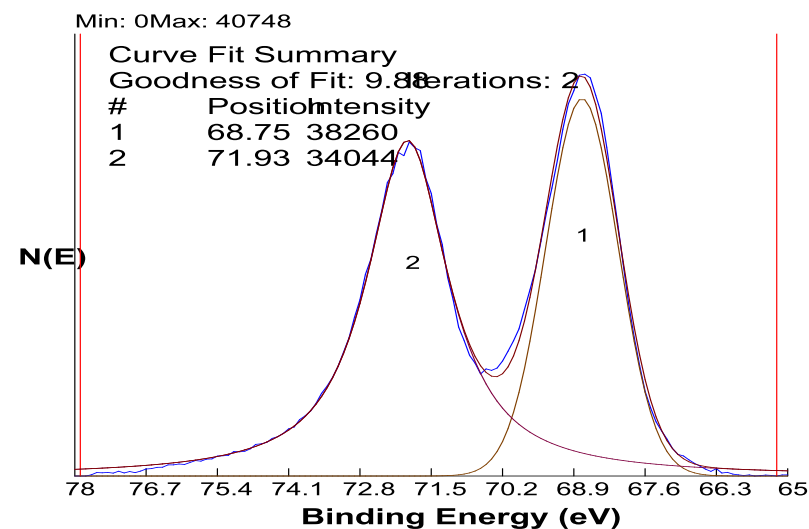


Pt 4f Peaks

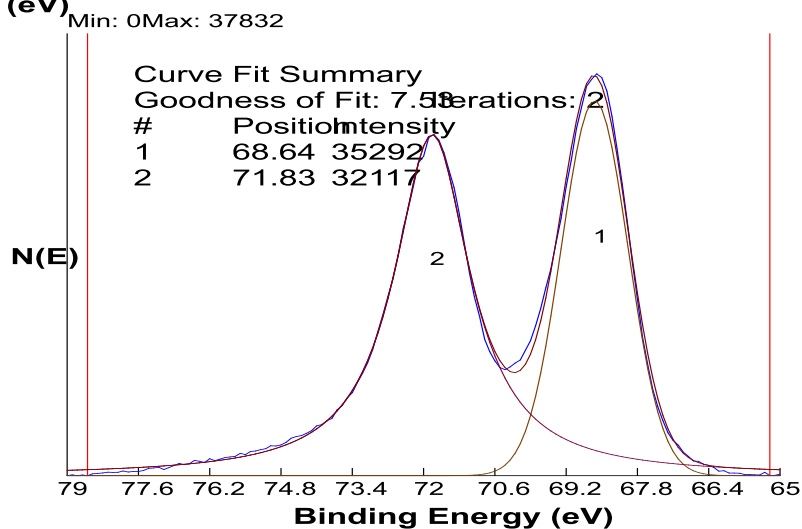
Curves:

1. Pt 4f<sub>7/2</sub>
2. Pt 4f<sub>5/2</sub>

Post-vacuum anneal



Vacuum  
anneal 300°C



# XPS Conclusions & Future Work

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## Conclusions:

- Oxynitride layer on AlGaN possible barrier for  $\text{SiN}_x/\text{AlGaN}$  reaction
- Gate metal/GaN chemical changes due to 300°C anneal in UHV:
  - Ni reacts with O from native oxide on GaN → material properties of NiO detrimental to Ni gate contact operation
  - Pt does not readily react with O from native oxide on GaN → better suited for gate contact operation

## Future Work:

- AlGaN UV ozone exposure followed by  $\text{SiN}_x$  passivation → chemical changes due to vacuum anneal
- Introduction of oxynitride layer on GaN before gate metal deposition → chemical changes due to vacuum anneal